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STUDIES OF COATINGS PRODUCED BY HIGH-VELOCITY OXY-FUEL SPRAYING USING CERMET POWDER BASED ON FeMoNiCrB AMORPHIZING ALLOY

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ABSTRACT

The process of producing composite powders based on amorphizing Fe-alloy with the additives of refractory compounds by the method of high-velocity oxy-fuel spraying was investigated. For spraying composite powders FeMoNiCrB-(Ti, Cr)C, FeMoNiCrB-ZrB₂ were used, produced from a mixture of powders of the compositions by mechanical alloying in a planetary mill. As a result of spraying, dense coatings (porosity is less than 3 %) were produced, which were formed from partially deformed particles with a multiphase structure and a uniform distribution of structural components. The results of X-ray diffraction phase analysis indicate the formation of amorphous-crystalline structure in the produced composite coatings. On the X-ray patterns, the maximum peak amplitude from the crystalline phase against the background of the amorphous halo corresponds to the TiCN phase in the coating FeMoNiCrB-(Ti, Cr)C and the ZrB₂ phase in the FeMoNiCrB-ZrB coating. The size of the measured microhardness for the composite coating FeMoNiCrB-(Ti, Cr)C amounts to — 5.5±0.25 GPa, and for the coating FeMoNiCrB-ZrB₂ it is 5.9±0.29 GPa.

KEY WORDS: high-velocity oxy-fuel spraying, amorphous phase, amorphous iron-based alloy, composite powder, composite coating, microstructure, microhardness

INTRODUCTION

Amorphous metallic iron-based materials are widely used in industry due to their advantages, such as high strength and hardness, excellent corrosion and wear resistance, good magnetic properties, as well as relatively low cost of material [1–3]. However, a disadvantage of amorphous compact materials is their low ductility and excessive brittleness at room temperature, as well as low efficiency of equipment and high production costs, which significantly limits their practical application as structural materials [4]. To eliminate these disadvantages, materials based on amorphous alloys are used on the surface of products in the form of protective coatings, which are produced by thermal spraying. The basis for the scientific and practical interest of the use of amorphous iron-based coatings and amorphous composite coatings in order to increase the stability of the surface of products is the cooling rate in thermal spraying of the powder melt particles, which is 10⁵–10⁶ K/s and is sufficient for amorphization of the coating material. In addition, amorphous metal coatings can be applied on large-sized and complex parts. This allows expanding the scope of their practical application.

For deposition of coatings from amorphizing iron-based alloys, the methods of plasma, detonation, electric arc and high-velocity oxy-fuel (HVOF, HVOF) spraying are used. The produced coatings are

used to increase the corrosion stability of containers for storage and transportation of spent nuclear fuel as an alternative replacement of expensive nickel and titanium alloys [5]; on parts operating in the conditions of corrosion, corrosion-erosion and abrasive wear for replacement of galvanic chrome plating [6, 7]; on parts of mobile warehouses [8]; to strengthen and restore the pipes of industrial boilers, operating in the conditions of high-temperature erosion wear [9]. The process of HVOF spraying is most widely used to produce coatings with an amorphous structure due to the use of a relatively low temperature and high jet velocity in this method, which results in the formation of a coating with a high density and adhesion strength with the base.

Along with the use of iron-based alloys, composite coatings with an amorphous matrix are developed, which was strengthened by the second phase. As reinforcing additives, TiN nitrides, B₄C and WC carbides, CrB₂ borides, Al₂O₃ oxides, as well as stainless steel, NiCr, etc. are used [10–14]. These composite coatings show a significant increase in hardness, wear resistance and corrosion resistance as compared to the basic analogue.

In this paper, a study of the structure formation and phase composition of composite coatings, produced by HVOF based on Fe-alloy, which is amorphised, with reinforcing additives of refractory compounds (Ti, Cr)C, ZrB₂, was carried out.

Table 1. Characteristics of powders for oxy-fuel spraying of coatings

Composition, wt.%	Particle size, μm	Method of producing
FeMoNiCrB (36.2Fe–29.9Mo–23.6Ni– 7.6Cr–2.7B)	<40	Spraying of melt by nitrogen
77 FeMoNiCrB–23 (Ti, Cr)C	<40	MA in PM 1.5 h
75 FeMoNiCrB–25 ZrB ₂	<80	

MATERIALS AND PROCEDURES OF INVESTIGATIONS

For HVOF coatings with an amorphous structure, powders based on amorphous alloy FeMoNiCrB were used, produced by mechanical alloying (MA) in a planetary mill (PM) [15]. The characteristics of powders are shown in Table 1.

The spraying was performed in the installation (HVOF) UVShGPN-M1 on the following technological parameters of the process: propane-butane pressure is 4 atm, oxygen pressure is 7 atm, air pressure is 6 atm, nitrogen pressure is 5 atm, spraying distance is 120 mm. The coatings FeMoNiCrB–(Ti, Cr)C and FeMoNiCrB–ZrB₂ were deposited to the substrate NiCr (thickness is 50–100 μm), which was sprayed by electric arc method (wire diameter is 2 mm).

In metallographic examinations, an optical microscope Neophot-32 with a digital photography device was used; measurement of microhardness was carried out in a PMT-3 device. X-ray diffraction analysis (XRD) of coatings was carried out in the installation DRON-UM-1, CuK α radiation, monochromatic.

RESULTS OF INVESTIGATIONS

Metallographic analysis revealed that coatings produced from all the studied materials have a dense,

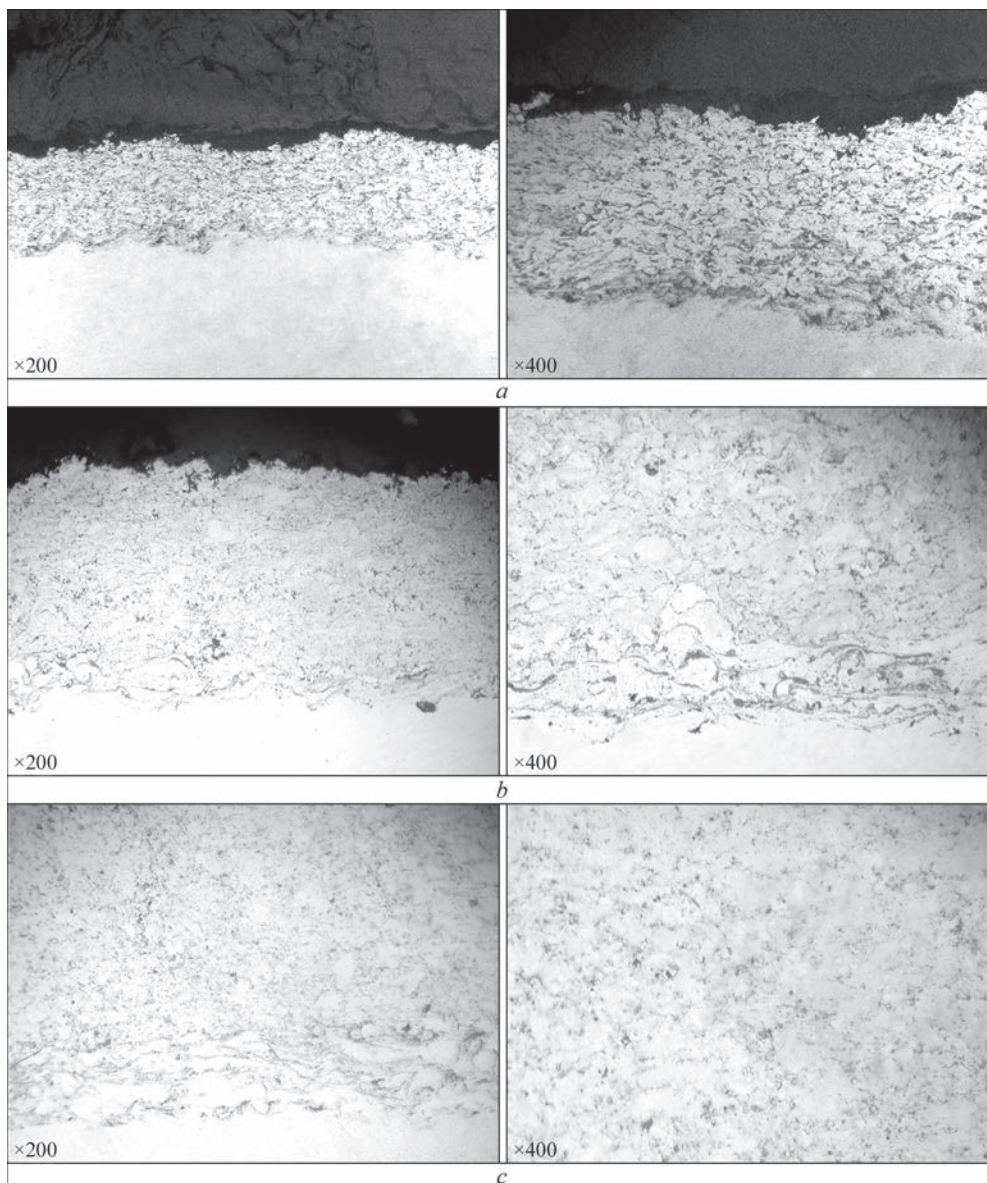
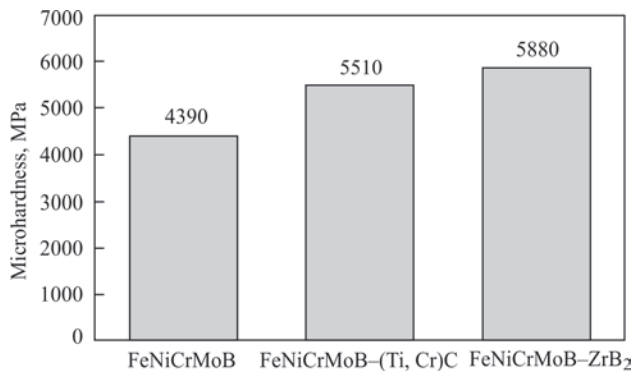


Figure 1. Microstructure of HVOF-coatings: *a* — FeMoNiCrB; *b* — FeMoNiCrB–(Ti, Cr)C; *c* — FeMoNiCrB–ZrB₂

Table 2. Results of investigations of HVOF-coatings based on FeMoNiCrB

Coating material	Thickness, μm	Microhardness $HV_{0.05}$, MPa	Phase composition
FeMoNiCrB	200–250	4390±290	A Φ ; Mo ₂ FeB ₂ ; Fe ₂ B; FeCr; Fe ₃ O ₄ ; Cr ₇ Ni ₃
FeMoNiCrB–(Ti, Cr)C	300–350	5510±250	A Φ ; Fe ₂ Ti; Cr ₇ Ni ₃ ; TiC _{0.3} N _{0.7} ; TiC _{0.2} N _{0.8} ; Fe _{15.1} C; Ni ₃ Fe; phases in small quantities: Ni ₃ C; MoC; FeMoO ₄ ; FeO; FeC ₈
FeMoNiCrB–ZrB ₂	900–950	5880±290	A Φ ; ZrB ₂ ; Fe; (Fe, Ni) solid solution; Cr ₇ Ni ₃ ; MoNi ₄ ; Ni ₂ Zr; phases in small quantities: ZrO ₂ ; Fe ₂ B; Ni ₂ B; MoB ₂ ; FeO; Ni _{0.4} Fe _{2.6} O ₄

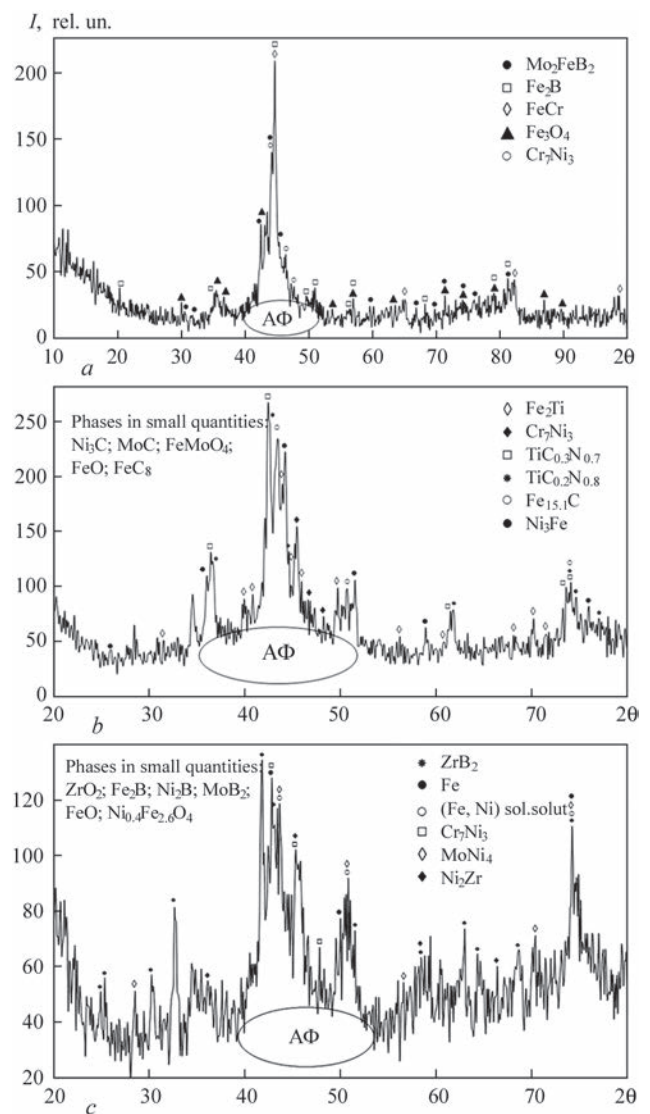
**Figure 2.** Microhardness of coatings with an amorphised structure produced by the method of HVOF-spraying

fine-grained structure, homogeneous over the whole structure area, which is formed from partially deformed spherical particles (Figure 1). Porosity of the coatings does not exceed 3 vol.%; the coatings FeMoNiCrB–(Ti, Cr)C and FeMoNiCrB–ZrB₂ tightly adhere to the nichrome sublayer and the coating FeMoNiCrB — to the steel base.

Measurement of microhardness of HVOF coatings (Figure 2) showed that the use of reinforcing additives (Ti, Cr)C and ZrB₂ leads to an increase in microhardness of composite coatings by 1120 and 1490 MPa as compared to the coating of the initial FeMoNiCrB powder.

X-ray diffraction analysis revealed (Figure 3) that in the investigated HVOF-coatings, the amorphous phase (A Φ) is present.

Data from XRD indicate that as a result of HVOF-spraying of powders based on FeMoNiCrB alloy, multiphase coatings were produced, having an amorphous crystalline structure. All the coatings have additional crystalline peaks against the background of halo from the amorphous phase. On X-ray patterns, the maximum peak on the amplitude from the crystalline phase against the background of the amorphous halo corresponds to the phases of Fe₂B and FeCr in the coating FeMoNiCrB; TiC_{0.3}N_{0.7} in the coating FeMoNiCrB–(Ti, Cr)C and the phase ZrB₂, in the coating FeMoNiCrB–ZrB₂. In all the coatings, the intermetallide phase Cr₇Ni₃ was recorded, having a tetragonal crystalline structure. Iron, as the main ele-

**Figure 3.** X-ray patterns of HVOF-coatings: *a* — FeMoNiCrB; *b* — FeMoNiCrB–(Ti, Cr)C; *c* — FeMoNiCrB–ZrB₂

ment of the initial alloy, is presented in the coatings in the form of oxides (Fe₃O₄, FeO, FeMoO₄, Ni_{0.4}Fe_{2.6}O₄), borides (Mo₂FeB₂, Fe₂B), intermetallides (FeCr, Fe₂Ti, Ni₃Fe), carbides (Fe_{15.1}C, FeC₈) and (Fe, Ni)-solid solution. In the coating FeMoNiCrB–ZrB₂, iron in a pure form was also revealed.

The obtained results of the study of HVOF-coatings based on the FeMoNiCrB alloy are shown in Table 2.

CONCLUSIONS

Applying the method of high-velocity oxy-fuel spraying using composite powders produced by mechanical alloying based on Fe-alloy that is amorphised, the coatings FeMoNiCrB-(Ti, Cr)C and FeMoNiCrB-ZrB₂ with an amorphous crystalline structure were produced.

The produced coatings are characterized by a uniform distribution of structural components, have a structure, that is homogeneous over the entire area, which is formed from partially deformed spherical particles. The porosity of coatings does not exceed 3 %.

It was revealed that the presence of reinforcing components (Ti, Cr)C, ZrB₂ leads to an increase in microhardness of coatings as compared to the coating of the initial FeMoNiCrB powder by 1120 and 1490 MPa, respectively.

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CONFLICT OF INTEREST

The Authors declare no conflict of interest

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